RESEARCH SYNOPSIS

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Summary. I am trained as an experimental materials scientist, with skills in the domain of materials characterization on electrochemical energy storage systems and ceramic/cement-based binder materials. Additionally, I utilize techniques such as chemical vapor deposition, sol-gel synthesis, and steric entrapment for synthesizing carbon films and aluminosilicate materials relevant for my research. I also develop and maintain my own Python scripts for rapid data analysis and plotting.

My doctoral research focuses on correlating carbon structures with performance as electrodes for Li-ion, Na-ion and redox-flow batteries. Additionally, I have industry experience through an internship, where I worked on developing Li metal batteries with solid-state polymer electrolytes. My master's thesis focused on identifying nucleation mechanisms in novel aluminosilicate binders intended as a replacement for cement in construction applications. With the exposure to these classes of materials, I can apply my skills towards practical problems in batteries, ceramic and cement materials, which constitute some of the most vital technological systems in the modern world.

Doctoral Research. The 2019 Nobel Prize in Chemistry for the development of Li-ion batteries serves to highlight its impact on modern lifestyle. However, in the bigger picture, grand challenges exist in the development of electrochemical energy storage solutions for applications such as grid-level energy storage.[1] With current lithium reserves being insufficient for such a huge demand, additional low-cost and versatile battery technologies are needed, prepared with earth-abundant materials such as carbon.

My doctoral research focuses on a two-pronged approach for i) developing analytical techniques and ii) tailoring interfaces on carbon materials, to advance the understanding of electrode-electrolyte interfaces in Li-ion and beyond Li-ion battery technologies.

a) Scanning Electrochemical Microscopy (SECM) for characterizing flow battery interfaces

Non-aqueous redox flow batteries (nRFBs) are an upcoming candidate for grid-level storage applications,[1] with energy being stored by charging redox-active organic molecules (ROMs). These high-energy molecules can potentially take part in parasitic processes at the interface

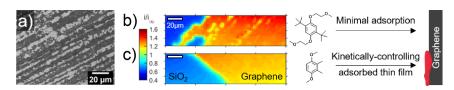


Figure 1: SECM characterization of electron-transfer (ET) rates between graphene films and ROMs. a) SEM micrograph of graphene showing variation in layer numbers of graphene. b) and c) SECM measurements revealing spatial distribution in electrochemical reactivity, with color bar quantifying ET kinetics. The lack of heterogeneity in ET kinetics due to varying layer numbers of graphene in c) is attributed to the formation of films.[2]

during (dis)charge, thereby decreasing the capacity and power capabilities of an nRFB. My research led to the development of robust, widely applicable testing protocols with scanning electrochemical microscopy (SECM) to screen ROMs for parasitic reactions at electrode interfaces.[2]

In brief, SECM is an electrochemical scanning probe measurement that can spatially resolve and quantify electron transfer (ET) rate constants between a ROM and an electrode. [3, 4] Our results on practically relevant ROMs revealed inhibition of ET owing to film formation on basal planes of graphene electrodes (**Fig.** 1). Edge planes of carbon appeared to be immune to these film formation processes, therefore providing powerful design guidelines for carbon electrode material selection in nRFBs.

Further improvements in this approach were made to reduce the testing time, and extract parameters quantifying ET kinetics. To this end, I have developed a SECM-based point measurement using COMSOL finite element simulations. The method is capable of rapidly evaluating ET kinetics (k_f) between ROMs and any electrode material as a function of overpotential or polarization applied at the electrode (**Fig. 2a**).

Consequently, this approach enables comparison with theoretical kinetic models such Butler-Volmer, to check for deviations from ideal ET processes at the interface. Using this SECM approach, I have characterized the role carbon surface structure on ET kinetics. Single-layer graphene (SLG) electrodes were microfabricated to have with sp^3 hybridized

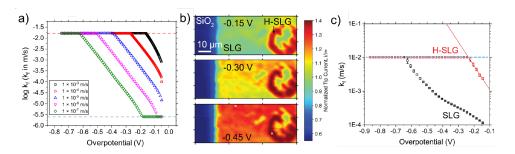


Figure 2: Quantifying ET behavior using COMSOL finite element simulations and experiments using SECM. a) Results from COMSOL simulations. b) SECM imaging revealing faster ET kinetics over H-SLG compared to SLG. c) SECM point measurement reveal ET kinetics to be conforming with Butler-Volmer model over H-SLG

hydrogenated graphene (H-SLG) using H_2 plasma. Ferrocene, an ideal ROM was observed to exhibit fast, ideal behavior over H-SLG, compared to pristine SLG, indicating molecular adsorption to be limiting ET kinetics (**Fig. 2b and 2c**). This phenomenon ties in closely to our previous observation of edge planes (likely functionalized and possessing sp³ hybridization) exhibiting no passivation in ET kinetics with practical ROMs for nRFBs.

b) Tailoring anode electrolyte interfaces in alkali-ion batteries through surface functionalization

Na-ion batteries are one of the promising and cheaper candidates in the realm of beyond Li-ion chemistries.[5] replacing Li with Na is However, with a major obstacle challenging, being the thermodynamic instability of Na⁺ intercalated graphite structures, among others. We hypothesized that these thermodynamic instabilities could be overturned by using 2D materials with distinct surface electronic properties that create favorable conditions for Na⁺ As a solution to this intercalation. problem, we recently presented theoretical calculations [6] which highlighted a fluorine surface modifier built over

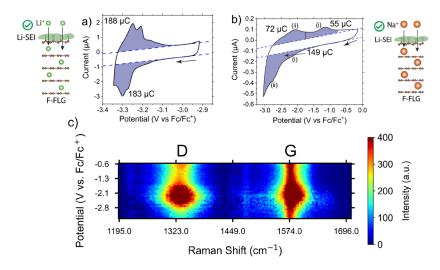


Figure 3: Na⁺ intercalation in fluorinated few-layer graphene (F-FLG). a) Initial SEI formation and measurement of Li⁺ capacity in F-FLG using CV at a rate of 1 mV/s. b) Subsequent Na⁺ intercalation signature in the same sample at 1 mV/s. c) In-situ Raman data showing intercalation signature for dilute Na⁺ intercalation in F-FLG, as evidenced from D and G peak broadening, and small shifts in D peak position

a thin, 4-layer graphene structure resulted in thermodynamically favorable Na⁺ intercalated graphene structures. Experimentally, using a combination of cyclic voltammetry (CV, **Fig. 3a, 3b**) and in-situ Raman spectroscopy (**Fig. 3c**), we were able to prove Na⁺ intercalation in these materials, with capacities ≈ 10 x higher than previously reported in experimental literature.[7] A key finding from our experimental study was the identification that the native solid-electrolyte interphase (SEI) formed on graphite anodes posed a

barrier to Na⁺ intercalation. We circumvented this problem by preforming a Li-based SEI by cycling F-FLG in Li-electrolyte solution, followed by cycling in Na-electrolyte solution to achieve Na⁺ intercalation. These results highlight surface modification as a potential strategy for enabling charge storage in thermodynamically unfavorable systems.

In addition to thermodynamic effects, we explored the effects of surface modification on the kinetics of Li⁺ intercalation processes, which in turn is heavily influenced by SEI formation. of multi-layer graphene (MLG) were modified by exposure to reactive plasma sources such as O₂ to yield O-MLG. Li⁺ intercalation was studied through CV and electrochemical impedance spectroscopy (EIS). CV measurements before and after cycling revealed greater deterioration in Li⁺ intercalation kinetics with MLG as compared to O-MLG anodes (Fig. 4a, 4b). A comprehensive EIS analysis accompanied by curve-fitting revealed that the sluggish kinetics were arising from Li⁺ transfer through the SEI (Fig. 4c). Ex-situ analysis using x-ray photoelectron spectroscopy (XPS) revealed significant differences in SEI chemical composition, notably in LiF content (Fig. 4d). These observations illustrate that surface functionalization routes could be used to direct SEI formation, resulting in a robust and functional interphase for fast Li⁺ kinetics in graphitic carbon anodes.

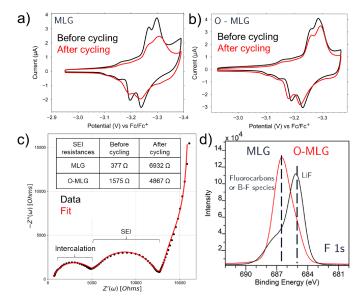


Figure 4: Li⁺ intercalation explored in surface modified MLG anodes). CV illustrating poor intercalation kinetics in a) MLG, compared to b) O-MLG anodes, as evident from peak splitting values. c) EIS data, and fit results revealing kinetic losses originating from mass transfer resistances across SEI. d) Ex-situ XPS of the SEI formed in MLG vs. O-MLG.

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